

## Hybrid Materials

International Edition: DOI: 10.1002/anie.201914494  
German Edition: DOI: 10.1002/ange.201914494Ping-Pong Energy Transfer in Covalently Linked Porphyrin-MoS<sub>2</sub> ArchitecturesRuben Canton-Vitoria<sup>+</sup>, Tobias Scharl<sup>+</sup>, Anastasios Stergiou, Alejandro Cadranell, Raul Arenal,\*  
Dirk M. Guldi,\* and Nikos Tagmatarchis\*

**Abstract:** Molybdenum disulfide nanosheets covalently modified with porphyrin were prepared and fully characterized. Neither the porphyrin absorption nor its fluorescence was notably affected by covalent linkage to MoS<sub>2</sub>. The use of transient absorption spectroscopy showed that a complex ping-pong energy-transfer mechanism, namely from the porphyrin to MoS<sub>2</sub> and back to the porphyrin, operated. This study reveals the potential of transition-metal dichalcogenides in photosensitization processes.

Molybdenum disulfide (MoS<sub>2</sub>) is a typical example of a layered transition-metal dichalcogenide. In general, the structure of such dichalcogenides is analogous to that of graphene, with an atomic layer of a transition metal sandwiched between two layers of chalcogen atoms. More specifically, each Mo atom is bound to six S atoms and forms a three-atom-thick monolayer.<sup>[1]</sup> Exfoliated MoS<sub>2</sub> materials have received enormous attention in recent years because of their extraordinary optoelectronic and electrocatalytic properties, especially in the area of energy applications and catalysis.<sup>[2,3]</sup> Diverse exfoliation strategies have been developed for MoS<sub>2</sub>. Most notable is the employment of a) strong intercalants such as Li<sup>+</sup>, which result in phase transfer to an octahedral metallic 1T polytype,<sup>[4]</sup> and b) non-oxidative Brønsted acids such as chlorosulfonic acid, which offer the benefit of retaining the 2H semiconducting phase.<sup>[5]</sup>

Additional efforts have been placed on developing strategies for chemically modifying transition-metal dichalcogenides.<sup>[6]</sup> The aim has been to fully harness the properties of, for example, MoS<sub>2</sub>. In fact, incorporation of molecular dopants results in fine-tuning of the electronic as well as optical properties of MoS<sub>2</sub> which, in turn, enables broadening the spectrum of its applications. In MoS<sub>2</sub>, the S atoms in the basal plane are, however, rather inert as a result of saturation. Furthermore, Mo atoms are placed between S layers and, thereby have challenging chemical reactivity. However, the covalent functionalization of MoS<sub>2</sub> has been accomplished.<sup>[7]</sup> For example, the basal plane of the 1T-MoS<sub>2</sub> polytype has been reacted with organoiodides,<sup>[8]</sup> while reaction with diazonium salts has been reported for the basal plane modification of 1T<sup>[9]</sup> as well as 2H-MoS<sub>2</sub>.<sup>[10]</sup> Herein we adopt the recently developed method based on the addition of 1,2-dithiolanes to S-vacant sites located at the edges of exfoliated 2H-MoS<sub>2</sub> nanosheets.<sup>[11]</sup> A true advantage of this approach is that diverse 1,2-dithiolanes are easily prepared and they react with exfoliated MoS<sub>2</sub> to yield interesting hybrids. Pyrene and phthalocyanines have, for example, been coupled with MoS<sub>2</sub>.<sup>[11]</sup> Alternatively, ammonium moieties have been grafted onto MoS<sub>2</sub> and WS<sub>2</sub>, which facilitated the electrostatic association of carbon dots<sup>[12]</sup> and anionic porphyrins.<sup>[13]</sup> In the resulting ensembles, excited-state electronic interactions between the components were responsible for quenching the fluorescence from the pyrene, carbon dots, or

[\*] Dr. R. Canton-Vitoria,<sup>[+]</sup> Dr. A. Stergiou, Prof. Dr. N. Tagmatarchis  
Theoretical and Physical Chemistry Institute  
National Hellenic Research Foundation  
48 Vassileos Constantinou Avenue, 11635 Athens (Greece)  
E-mail: tagmatar@eie.gr  
T. Scharl,<sup>[+]</sup> Dr. A. Cadranell, Prof. Dr. D. M. Guldi  
Department of Chemistry and Pharmacy & interdisciplinary Center  
for Molecular Materials (ICMM)  
Friedrich-Alexander Universität Erlangen-Nürnberg  
Egerlandstrasse 3, 91058 Erlangen (Germany)  
E-mail: dirk.guldi@fau.de

Dr. A. Cadranell  
Universidad de Buenos Aires  
Facultad de Ciencias Exactas y Naturales  
Departamento de Química Inorgánica, Analítica y Química Física  
Pabellón 2, Ciudad Universitaria  
C1428EHA Buenos Aires (Argentina)  
and  
CONICET—Universidad de Buenos Aires  
Instituto de Química-Física de Materiales  
Medio Ambiente y Energía (INQUIMAE)  
Pabellón 2, Ciudad Universitaria  
C1428EHA Buenos Aires (Argentina)

Dr. R. Arenal  
Laboratorio de Microscopias Avanzadas (LMA)  
Instituto de Nanociencia de Aragón (INA), U. Zaragoza  
Mariano Esquillor s/n, 50018 Zaragoza (Spain)  
and  
Instituto de Ciencias de Materiales de Aragón  
CSIC-U. de Zaragoza  
Calle Pedro Cerbuna 12, 50009 Zaragoza (Spain)  
and  
ARAID Foundation, 50018 Zaragoza (Spain)  
E-mail: arenal@unizar.es

[+] These authors contributed equally to this work.

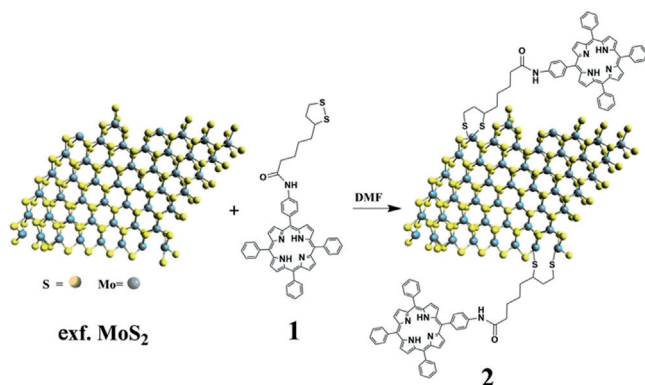
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porphyrin. The major drawbacks of electrostatically associated ensembles are their lack of stability and processability in organic solvents. They also have moderate binding strengths, especially when compared to the robust and strong bonding in covalently linked conjugates. Therefore, it is not only imperative, but also timely, to explore covalent modification of MoS<sub>2</sub> with photoactive porphyrins.

Herein, we report on the modification of the edges of exfoliated MoS<sub>2</sub> with a 1,2-dithiolane derivative **1** featuring a porphyrin (H<sub>2</sub>P). The newly prepared H<sub>2</sub>P-MoS<sub>2</sub> **2** was comprehensively characterized by spectroscopic, thermal, and microscopy means. We also gathered insight into the electronic interactions between the porphyrin and MoS<sub>2</sub> on the femto- to nanosecond timescales upon photoexcitation.

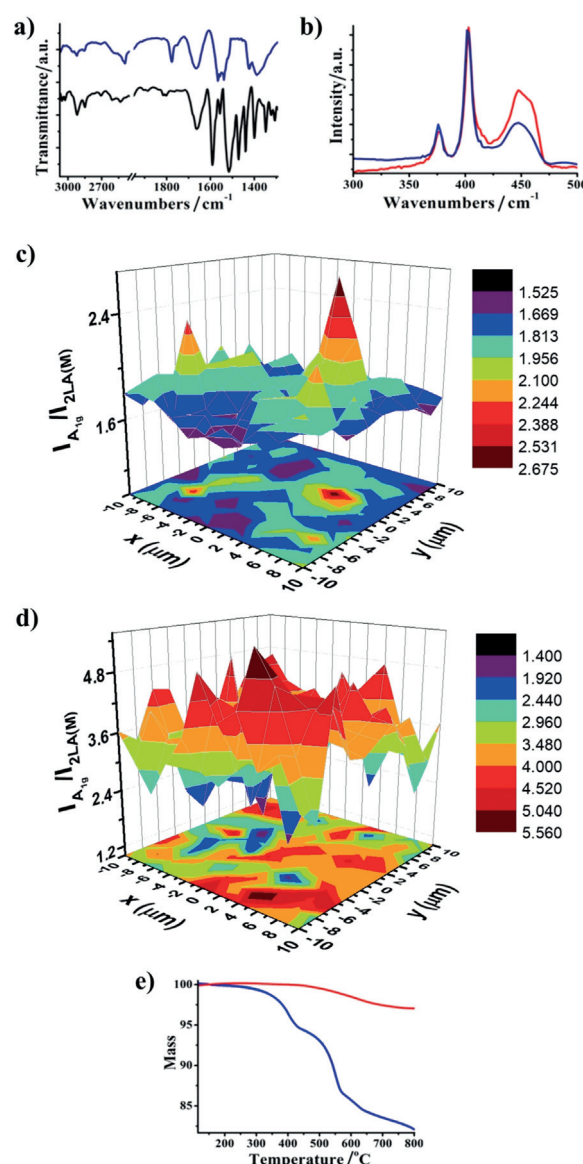
1,2-Dithiolane-based porphyrin **1** was synthesized by a condensation reaction between  $\alpha$ -lipoic acid and 5-(4-aminophenyl)-10,15,20-(triphenyl)porphyrin. In parallel, treatment of bulk MoS<sub>2</sub> with chlorosulfonic acid allowed exfoliation of semiconducting nanosheets.<sup>[5]</sup> Next, the reaction of exfoliated MoS<sub>2</sub> with **1** yielded H<sub>2</sub>P-MoS<sub>2</sub> **2**, as summarized in Figure 1. Filtration of the reaction mixture



**Figure 1.** Reaction route for obtaining H<sub>2</sub>P-MoS<sub>2</sub> **2**.

through a PTFE membrane (0.2  $\mu$ m pore size) followed by extensive washing with dichloromethane, assured the removal of any noncovalently physisorbed **1**, as revealed by UV/Vis spectroscopy (Figure S1). Purified **2** showed reasonable solubility in DMF, benzonitrile, and 2-propanol, while it is completely insoluble in dichloromethane and water.

Evidence for the success in the covalent modification of MoS<sub>2</sub> with H<sub>2</sub>P came from vibrational spectroscopy. Attenuated total reflectance infrared (ATR-IR) measurements on H<sub>2</sub>P-MoS<sub>2</sub> **2** (Figure 2a) revealed the presence of a) C–H stretching vibrations corresponding to the alkyl chain, which connects H<sub>2</sub>P with MoS<sub>2</sub>, at 2848 and 2950 cm<sup>−1</sup>, b) an amide carbonyl vibration at 1660 cm<sup>−1</sup>, and c) an aromatic C=C bending at 1590 cm<sup>−1</sup>. We further validated the origin of the IR bands related to H<sub>2</sub>P within the hybrid by performing a reference experiment in which a mixture of tetraphenylporphyrin and exfoliated MoS<sub>2</sub> was processed under the same experimental conditions as those for obtaining **2**. The presence of physisorbed tetraphenylporphyrin was not detected in the UV/Vis and emission spectra of the reference



**Figure 2.** a) ATR-IR spectra of **1** (black) and **2** (blue). b) Raman spectra of exfoliated MoS<sub>2</sub> (red) and **2** (blue) upon excitation at  $\lambda = 633$  nm. c, d) Raman mapping upon excitation at  $\lambda = 633$  nm of the  $I_{A_{1g}}/I_{2LA(M)}$  intensity ratio of a  $20 \times 20 \mu\text{m}^2$  area for c) exfoliated MoS<sub>2</sub> and d) **2**. e) TGA plots of exfoliated MoS<sub>2</sub> (red) and **2** (blue).

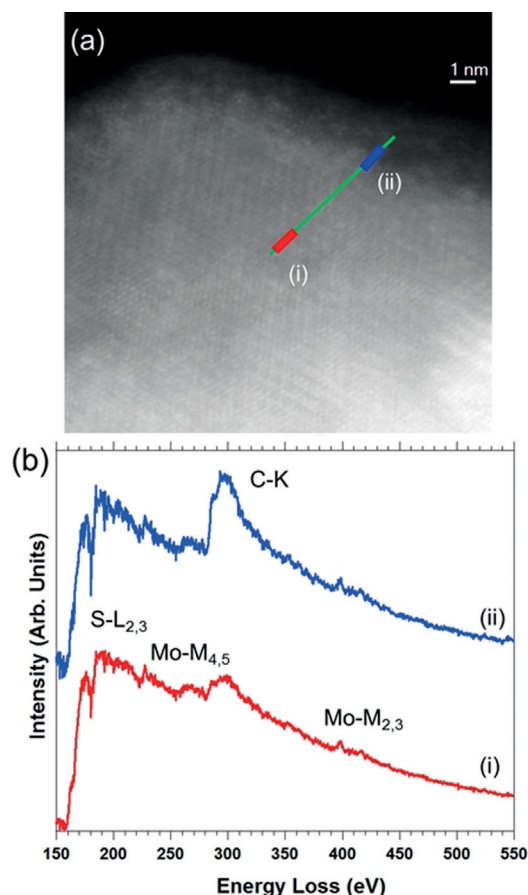
material, and no contribution in the mass loss detected by TGA was observed (Figures S2–S4).

Additional structural information regarding **2** came from Raman spectroscopy. Excitation at  $\lambda = 514$  nm showed the existence of the C=C and C=N bonds of H<sub>2</sub>P in the 1000–1600 cm<sup>−1</sup> region (Figure S5). Characteristic modes corresponding to MoS<sub>2</sub> were observed in the lower wavenumber region, especially upon excitation at  $\lambda = 633$  nm. These modes become visible as a result of coupling with the A<sub>1</sub> excitonic transition, which in turn produces resonance Raman enhancement of the first and second order vibrational modes. Exfoliated semiconducting MoS<sub>2</sub> gave rise to bands corresponding to 2LA(M) at 447 cm<sup>−1</sup>, A<sub>1g</sub> at 406 cm<sup>−1</sup>, and E<sub>2g</sub><sup>1</sup> at 382 cm<sup>−1</sup> (Figure 2b). The frequency difference between A<sub>1g</sub>

and  $E_{2g}^1$  was  $24\text{ cm}^{-1}$ , from which we conclude the existence of few-layered  $\text{MoS}_2$ .<sup>[14,15]</sup> The  $2\text{LA(M)}$  vibration relates to the S-vacancies.<sup>[16]</sup> By comparing the Raman spectrum of exfoliated  $\text{MoS}_2$  with that of **2**, the intensity ratio of  $A_{1g}$  compared to  $2\text{LA(M)}$  was found to be considerably higher for **2**, that is, 3.6 versus 1.9 for exfoliated  $\text{MoS}_2$ . Complementary Raman mapping assays revealed an increased  $I_{A_{1g}}/I_{2\text{LA(M)}}$  intensity ratio for **2** compared to that of exfoliated  $\text{MoS}_2$  (Figure 2c,d). This observation is in sharp contrast to the recorded  $I_{A_{1g}}/I_{2\text{LA(M)}}$  maps for the reference material prepared by physisorption (Figure S6). Overall, this is indicative of a “healing” of the S-vacancies by the 1,2-dithiolane species in **2**.<sup>[9]</sup> Likewise, the absence of features at  $150$ ,  $225$ , and  $325\text{ cm}^{-1}$ , indicative of the metallic polytype, highlights the semiconducting nature of  $\text{MoS}_2$  in **2**.

Thermogravimetric analysis (TGA) provided information related to the degree of  $\text{MoS}_2$  functionalization in **2**. Exfoliated  $\text{MoS}_2$  was found to be thermally stable under  $\text{N}_2$  from  $100$  to  $800^\circ\text{C}$ . Notably, the mass loss observed in the modified  $\text{MoS}_2$  is directly related to the thermal decomposition of the organic addends. Hence, the mass loss of 9% for **2** (Figure 2e), which occurred up to  $520^\circ\text{C}$ , is ascribed to the porphyrin. Based on this mass loss, a loading of one  $\text{H}_2\text{P}$  moiety per 54 units of  $\text{MoS}_2$  was calculated. This is in good agreement with values reported for similar functionalizations of  $\text{MoS}_2$  with pyrene and zinc phthalocyanine.<sup>[11]</sup>

Spatially resolved electron energy loss spectroscopy (SR-EELS) using a scanning transmission electron microscope (STEM) enables the morphology and chemical composition of complex hybrid nanostructures to be investigated at the local scale.<sup>[17]</sup> Thus, we performed SR-EELS STEM analyses of **2**. Figure 3 shows an EEL spectrum line (SPLI) of a functionalized  $\text{MoS}_2$  flake (see also Figure S7). Figure 3a corresponds to the high angular annular dark field (HAADF) STEM image of the flake and the SPLI was acquired along the highlighted green line. Typical EEL spectra are displayed in Figure 3b. Each of them corresponds to the sum of four EEL spectra extracted from the two areas marked in the SPLI (red (i) and green (ii) lines, respectively). S- $L_{2,3}$ , Mo- $M_{4,5}$ , C-K, and Mo- $M_{2,3}$  edges are observed in both spectra. The energy-loss near-edge structure (ELNES) investigations of the C-K edge are displayed in the right panel of Figure S7. Two signatures at  $291.2$  and  $293.4\text{ eV}$  can be observed in the  $\sigma^*$  region (ca.  $290\text{--}325\text{ eV}$ ) for **2**. Such signals are not discerned in the spectra of other carbon-containing materials employed as references. Some differences can also be observed in the main contribution to the  $\pi^*$  signal at  $\approx 284\text{--}290\text{ eV}$ . Here, the C-K edge is dominated by an aromatic C contribution (ca.  $285.5\text{ eV}$ <sup>[17b,c,18]</sup>), which is less pronounced in the case of  $\text{H}_2\text{P}$ . Furthermore, another feature at about  $287.2\text{ eV}$  is clearly seen in the spectra of **2**. This signal can be attributed to pyrrolic (C-N) contributions of  $\text{H}_2\text{P}$ .<sup>[18c]</sup> Importantly, the N-K edge cannot be observed in  $\text{H}_2\text{P}$  because the Mo- $M_{2,3}$  edge is in the same energy range. For all these reasons, the features highlighted in the C-K edge of **2** correspond to a porphyrin contribution. Furthermore, we conclude from the EELS analyses that  $\text{H}_2\text{P}$  is homogeneously distributed across the surface of the  $\text{MoS}_2$  flakes in **2** (Figure S8).

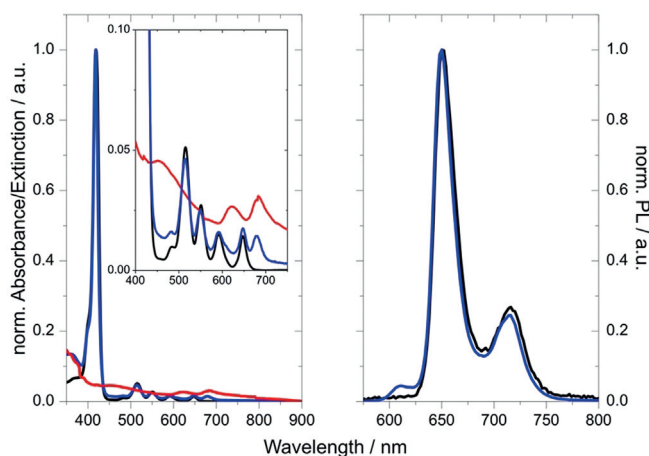


**Figure 3.** a) HAADF HRSTEM image of a flake of **2**. An EELS SPLI was acquired along the green line. b) Two EEL spectra, each corresponding to the sum of four selected EEL spectra collected from the two marked regions in the SPLI shown in (a). The S- $L_{2,3}$ , Mo- $M_{4,5}$ , C-K, and Mo- $M_{2,3}$  edges are visible in both spectra. The carbon detected is associated with  $\text{H}_2\text{P}$ .

Next, focusing on the optical properties of **2**, the UV/Vis spectrum was recorded in DMF and compared with that of **1**. In particular, the UV/Vis spectrum of **1** showed a sharp absorption at  $\lambda = 418\text{ nm}$  that corresponds to the Soret band, accompanied by weaker Q-band absorptions at  $\lambda = 516$ ,  $550$ ,  $596$ , and  $648\text{ nm}$  (Figure 4, left). Spectroscopic fingerprints at  $\lambda = 401$ ,  $494$  (C-exciton),  $626$  (B-exciton), and  $681\text{ nm}$  (A-exciton) are discernable in the UV/Vis spectrum of exfoliated  $\text{MoS}_2$  (Figure 4, left). The UV/Vis spectrum of **2** is best described as a superimposition of the individual spectra of **1** and exfoliated  $\text{MoS}_2$ . The spectrum is, for example, dominated by the characteristic Soret-band absorption of  $\text{H}_2\text{P}$  at  $\lambda = 418\text{ nm}$  in addition to the  $\text{MoS}_2$ -related transition at  $\lambda = 679\text{ nm}$ . From this similarity, we deduce that the ground-state lacks appreciable electronic interactions.

Excited-state electronic interactions could also be used to probe interactions between the  $\text{H}_2\text{P}$  and  $\text{MoS}_2$  in **2**. Specifically, strong fluorescence, centered at  $\lambda = 651$  and  $715\text{ nm}$  (Figure 4, right), was found upon excitation of **1** at the Soret-band absorption at  $\lambda = 420\text{ nm}$ . A fluorescent lifetime of  $10.3\text{ ns}$  (Figure S9) was determined by time-correlated single photon counting (TCSPC) measurements on **1** with DMF as

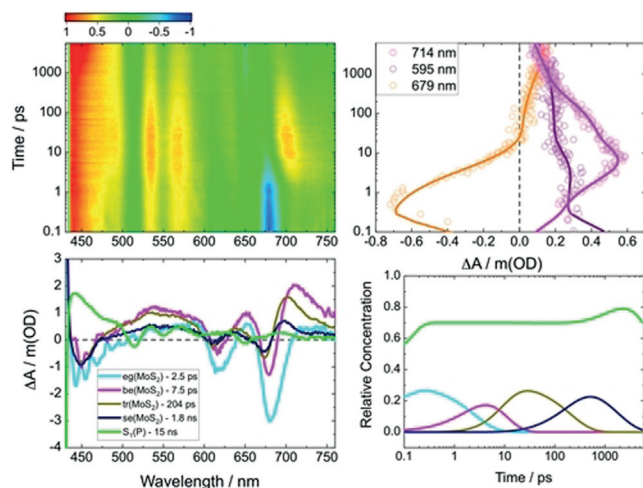




**Figure 4.** Left: Electronic absorption spectra of **1** (black), exfoliated MoS<sub>2</sub> (red), and **2** (blue) in DMF. Right: Fluorescence spectra of **1** (black) and **2** (blue) in DMF upon excitation at  $\lambda = 420$  nm.

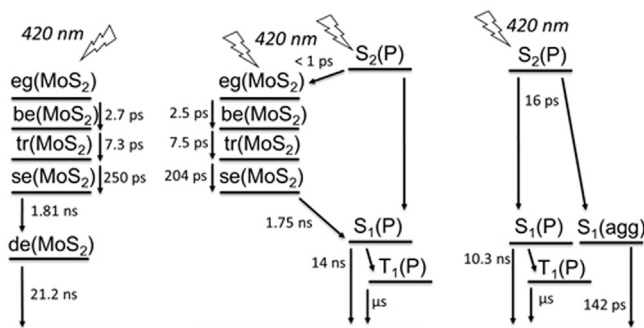
the solvent. Photoexcitation of **2** at  $\lambda = 420$  nm, however, resulted in no significant quenching of the porphyrin-centered fluorescence relative to that of **1** (Figure 4, right). For **2**, the fluorescence maxima were observed at  $\lambda = 649$  and  $714$  nm, and the fluorescence lifetime was  $10.3$  ns (Figure S9).

As a complement to the aforementioned method, we probed **1**, exfoliated MoS<sub>2</sub>, and **2** by means of femtosecond transient absorption spectroscopy (fsTAS) using an excitation wavelength of  $420$  nm (Figure 5, see also Figures S10 and S11). For **1**, bleaching of the ground state at  $\lambda = 517$  and  $653$  nm is observed early in the experiments. In addition, positive absorptions evolved at  $\lambda = 442$ ,  $538$ ,  $574$ ,  $623$ , and  $690$  nm (Figure S10). Global analysis revealed the presence of three exponential decays on the fsTAS timescale. These three



**Figure 5.** Top left: Differential absorption 3D map of **2** in DMF at room temperature with excitation at  $\lambda = 420$  nm. Top right: Time absorption profiles and fits at selected wavelengths. Bottom left: Species-associated differential spectra of exciton generation (e.g. [MoS<sub>2</sub>]: cyan curve), biexcitons (be[MoS<sub>2</sub>]: pink curve), trions (tr[MoS<sub>2</sub>]: brown curve), single excitons (se[MoS<sub>2</sub>]: dark blue curve), and singlet excited state (S<sub>1</sub>(H<sub>2</sub>P): green curve). Bottom right: Evolution of the different species over time.

decays are followed by a much slower decay, whose dynamics are outside the timescale of our experiments. Therefore, a target model was applied to fit the transient absorption data (Figure 6, right). This involves an initial population of a second singlet excited state [S<sub>2</sub>(H<sub>2</sub>P)], consistent with excitation into the Soret-band absorption, which internally



**Figure 6.** Deactivation models for MoS<sub>2</sub> (left), **2** (middle), and **1** (right) upon excitation at  $420$  nm.

converts into the first singlet excited state [S<sub>1</sub>(H<sub>2</sub>P)] within  $16$  ps. The latter undergoes intersystem crossing within  $10$  ns to the triplet state [T<sub>1</sub>(H<sub>2</sub>P)], which, in turn, decays back to the ground state on the microsecond timescale. Additionally, aggregates of **1** are present in the sample and they were found to decay to the ground state within  $142$  ps. These dynamics resemble very closely those already reported for a variety of porphyrins.<sup>[19,20]</sup>

For exfoliated MoS<sub>2</sub>, ground-state bleaching evolved at  $\lambda = 476$ ,  $623$ , and  $683$  nm together with positive absorptions at  $\lambda = 591$ ,  $655$ , and  $743$  nm (Figure S11). Global analysis of the data revealed five exponential decays after the absorption of light. The first three decays, which take place within  $2.7$ ,  $7.3$ , and  $250$  ps, involve biexciton [be(MoS<sub>2</sub>)] and trion [tr(MoS<sub>2</sub>)] formation as well as the decay of these many-body particles into single excitons [se(MoS<sub>2</sub>)]. Single excitons then diffuse within  $1.8$  ns across the layers in the  $z$  direction [de(MoS<sub>2</sub>)], before they recombine to recover the ground state within  $21.2$  ns (Figure 6, left). These dynamics resemble those already reported for exfoliated MoS<sub>2</sub>.<sup>[21]</sup>

For **2**, excited states of both components are observed to participate in the decay cascade (Figure 5). Selective excitation into the Soret-band absorption of **2** affords a photo-induced absorption at  $\lambda = 450$  nm and ground-state bleaching of the bands at  $\lambda = 617$  and  $679$  nm. Please note that, while the first absorption belongs to the porphyrin, the last two are fingerprints of MoS<sub>2</sub>. Considering that porphyrin absorptions in **2** are around  $20$  times stronger than those of MoS<sub>2</sub> at the excitation wavelength (Figure 4, left), the similar intensities of the differential features at  $\lambda = 450$  and  $679$  nm imply an ultrafast energy transfer ( $< 1$  ps) from S<sub>2</sub>, which is populated upon excitation at  $\lambda = 420$  nm, to MoS<sub>2</sub>. However, since the energy transfer is faster than the time resolution of our instrument, we are unable to measure the population of the S<sub>2</sub>(H<sub>2</sub>P). Global analysis showed four exponential decays on the fsTAS timescale and an additional decay, which is,

however, outside of our detection range. Our interpretation is based on a kinetic target model, which is depicted in the middle panel of Figure 6. This considers an initial excited state population governed by a porphyrin-centered  $S_2$ , with a minor contribution from a  $MoS_2$ -centered excited state. A nonquantitative ultrafast energy transfer from  $H_2P$  to  $MoS_2$  then occurs, which results in comparable populations of the  $H_2P$ -centered (70%) and  $MoS_2$ -centered (30%) excited states. Afterwards, these three species, which govern the deactivation of exfoliated  $MoS_2$  and which result in the formation of single excitons, are discernible. The times for the interconversions are 2.5, 7.5, and 204 ps. Once populated, the single exciton of  $MoS_2$  is subject to a second energy transfer. This time, however, single excitons are transferred from the  $MoS_2$  to the  $S_1$  state of  $H_2P$ . This energy transfer takes 1.75 ns. From here, the porphyrin decay proceeds through intersystem crossing to the corresponding triplet excited state within 14 ns, and recovery of the ground state on the microsecond timescale. This mechanism is also based on the energetics of **2**. The  $S_1$  state of **1**, with its fluorescence at 1.73 eV ( $\lambda = 715$  nm) and the  $MoS_2$  fluorescence at 1.82 eV,<sup>[22]</sup> render the second energy transfer a downhill process. This ping-pong energy-transfer model enables us to rationalize the nearly identical steady-state photophysical characteristics of **1** and **2** upon photoexcitation.

In summary, we accomplished the covalent grafting of porphyrin **1** onto exfoliated  $MoS_2$  to afford novel  $H_2P$ - $MoS_2$  **2**, which was fully characterized. Thorough photophysical investigations based on steady-state and time-resolved measurements corroborate that the decay of the photoexcited porphyrin involves a ping-pong energy transfer to and from  $MoS_2$ . Our findings suggest that transition-metal dichalcogenides have great potential in photosensitization. Such hybrid materials may be useful in energy-conversion applications.

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## Conflict of interest

The authors declare no conflict of interest.

**Keywords:** 1,2-dithiolane · hybrid materials ·  $MoS_2$  · porphyrins · sensitizers

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